THE STABILIZATION OF LiAsF₆/1,3-DIOXOLANE FOR USE IN RECHARGEABLE LITHIUM BATTERIES

J. L. GOLDMAN, L. A. DOMINEY and V. R. KOCH Covalent Associates, Inc., Woburn, MA 01801 (U.S.A.)

Summary

The use of highly conductive $\text{LiAsF}_{6}/1,3$ -dioxolane (DIOX) electrolytes in rechargeable Li batteries is precluded by the ease with which this electrolyte undergoes polymerization. By adding 0.5 - 1.0 vol.% 2-methylfuran (2-MeF) to $\text{LiAsF}_{6}/\text{DIOX}$ electrolytes, excellent shelf and cycle life may be obtained in Li/TiS_{2} cells. Stability tests indicate that 2-MeF acts on the TiS_{2} cathode and the bulk electrolyte as well as on the Li anode.

Introduction

One of the best electrolytes reported on to date for use in ambient temperature rechargeable Li batteries is the LiClO₄/DIOX system developed by Exxon in concert with their TiS₂ cathode [1]. Figures of Merit (FOM) in excess of 40 were reported for cycling from 40 - 60 C cm⁻² Li at discharge current densities of 2.5 mA cm⁻² [2]. Unfortunately, the propensity of this electrolyte to detonate both on impact and overdischarge precluded further work on this system by the Exxon group [3]. Efforts to cycle Li/TiS₂ cells by substituting LiPF₆ [3], LiAsF₆ [4], and either LiB(Ar)₄ or LiBR₄ [5] for LiClO₄ have largely failed. For the MF₆⁻ anions, DIOX polymerization was the rule. For the tetraalkyl and phenylborates, the best reported FOM for the Li electrode was 13 [5].

As part of our ongoing electrolyte development program we have considered substituting stabilized $\text{LiAsF}_6/\text{DIOX}$ electrolyte for $\text{LiAsF}_6/\text{tetra-hydrofuran}$ (THF) and/or 2-methyltetrahydrofuran (2-MeTHF) in Li/TiS_2 cells. $\text{LiAsF}_6/\text{DIOX}$ by itself, and blended with other ethers, may have the potential to provide better conductivities and lower temperature performance than current LiAsF_6^- -based ether electrolytes.

While a variety of stabilizers for LiAsF₆/DIOX electrolytes has been proposed [6], no work on the interaction of 2-methylfuran(2-MeF) with the components of the Li/LiAsF₆-DIOX/TiS₂ battery has been presented in the open literature. Because 2-MeF is able to extend greatly the cycling in tetrahydrofuran (THF) and 2-methyltetrahydrofuran (2-MeTHF)/LiAsF₆ systems [7], we have studied the chemical stability of the LiAsF₆/DIOX/2-MeF elec-

0378-7753/89/\$3.50

trolytes via 70 °C storage tests, and have cycled this electrolyte containing the 2-MeF additive in full Li/TiS_2 cells.

Experimental

1,3-Dioxolane (DIOX) (>99.98% purity) was obtained from Grant Chemical (Baton Rouge, LA) stabilized with 10 ppm of a proprietary preservative. DIOX and THF (Fisher HPLC grade) were stirred over Li/Hg amalgam in a Vacuum Atmospheres Corp. glove box and filtered through 0.7 μ m Whatman GF/F glass filters to remove reduced impurities. Lithium hexafluoroarsenate (LiAsF₆) (LaRoche Industries, Inc., Decatur, GA) was added to DIOX at 0 °C with stirring. 2-Methylfuran (2-MeF) (Aldrich) was distilled over CaH₂ and passed through alumina (Fisher, neutral Activity 1) prior to use.

Samples of DIOX and other cell components were incubated in Teflonlined, screw-cap culture tubes (Corning, C9826) at 70 ± 2 °C. Visual observations of electrolyte condition were made on a day-to-day basis.

Li/TiS₂ cells were assembled as previously reported [8]. The TiS₂ (Degussa, battery grade) was mixed with 8% Shawinigan carbon, 50% compressed and 8% Teflon which was then pressed onto expanded Ni screen (Delker) at 8000 psi. The cells were cathode limited to 260 mA h capacity and contained a Li/TiS₂ ratio of 3.8 to 1. The electrolyte consisted of 1.5 M LiAsF₆/DIOX containing 0.5 vol.% 2-MeF. The 20 cm² electrode area prismatic test cells were vacuum backfilled with 10 ml of electrolyte. We cycled the cells at 40% depth of discharge between 2.8 and 1.7 V at current densities of 1.0 mA cm⁻² on charge and discharge.

Results and discussion

Cycling studies

As a control, two cells were prepared without 2-MeF. Upon filling the control cells, the resistances climbed from the usual 2 - 3 Ω to over 30 Ω within 30 min of filling. Owing to the high resistance these cells were uncycleable and, upon opening, contained solid polymerized electrolyte. However, the best data obtained for cells containing 0.5% 2-MeF reached 155 cycles before cathode utilization fell below 40%, as shown in Fig. 1. The minimum Li FOM (Li turnover number) was found to be 18.9. By comparison, the 1.5 M LiAsF₆/THF electrolyte with, and without, 0.5% 2-MeF reaches 90 - 100 cycles (FOM ~14) and 15 - 20 cycles (FOM ~2.5), respectively [9]. Thus, the addition of 2-MeF to DIOX yields better cycling performance than when added to THF.

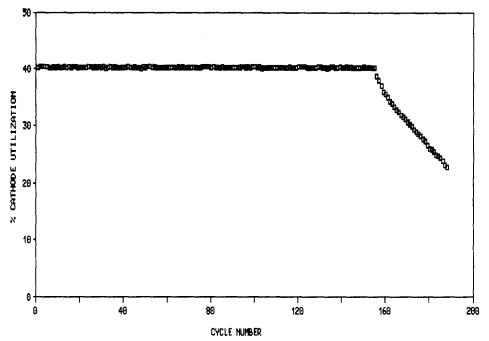


Fig. 1. Cathode utilization vs. cycle number for an Li/TiS₂ cell containing 1.5 M LiAsF₆/dioxolane + 0.5% 2-MeF.

TABL	E 1	
DIOX	stability	stud y ^a

Components	Observations	# Days	Temp. (°C)
DIOX	clear, colorless, no sign of polymerization	60	70
DIOX/LiAsF ₆	clear, colorless, viscous	30	25
	clear, colorless, polymeric gel	60	25
DIOX/LiAsF ₆	clear, colorless, viscous	1	70
	clear, colorless, solid polymer	2	70
DIOX/TiS ₂	clear, colorless, solid polymer	<1	25
$DIOX/LiAsF_6$ + TiS ₂	clear, colorless, solid polymer	<1	25
DIOX/LiAsF ₆	clear, light yellow, no polymer	4	70
+ 2-MeF	opaque, dark red, viscous	8	70
	opaque, dark red, solid polymer	5 9	70
$\frac{\text{DIOX}/\text{LiAsF}_6}{\text{+ 2-MeF} + \text{TiS}_2}$	clear, yellow, no polymer	1	70
	opaque, dark red, viscous	4	70
	opaque, dark red, solid polymer	29	70

^aNo Li present.

Storage tests

Table 1 summarizes the results of the DIOX stability study. Each tube contained 2 ml of either purified DIOX or 1.5 M $\text{LiAsF}_6/\text{DIOX}$ with, or without, 1 vol% of 2-MeF. Several tubes also contained 0.1 g of TiS₂.

The results indicate that not only does TiS_2 initiate the polymerization of DIOX but also $LiAsF_6$ does too. The addition of 2-MeF to help reduce DIOX polymerization indicated that 2-MeF may contribute to increased Li cycling by effects other than simple anode film formation. We believe that the formation of catalytically active species as a result of the electrolyte's intrinsic instability indicates that no amount of purification will ever eliminate all chemical degradation pathways. Accordingly, high cycle life in lithium cells will be dependent on additives which intercept reactive species in situ. In this regard we are currently working on other additives that, in concert with 2-MeF, provide more highly cycleable $LiAsF_6$ /DIOX electrolytes.

Conclusions

The work to date on 1.5 M LiAsF $_6$ /DIOX stabilized with 0.5 - 1.0 vol.% 2-MeF shows:

• TiS_2 , or impurities therein, can initiate rapid polymerization of DIOX while the presence of $LiAsF_6$ results in a slower polymerization rate. These findings confirm earlier studies in DIOX-based electrolytes lacking 2-MeF [4, 6].

• 2-MeF stabilizes the electrolyte by scavenging reactive impurities and/ or electrode/electrolyte reaction products. 2-MeF enhances Li cycling efficiency by effects other than simple anode film formation.

• Li FOMs in excess of 18 may be obtained by cycling full Li/TiS_2 cells with 2-MeF stabilized electrolyte.

Acknowledgement

This work was supported by the National Aeronautics and Space Administration.

References

- (a) G. H. Newman and L. H. Gaines, Ger. Offen., 26,19,806 (1976); (b) M. S. Whittingham and G. H. Newman, U. S. Pat. 4,086,403 (1978); (c) L. H. Gaines, U. S. Pat. 4,091,191 (1978).
- 2 K. M. Abraham and S. B. Brummer, in J. P. Gabano (ed.), Lithium Batteries, Academic Press, London, 1983, Ch. 14 and references therein.
- 3 G. H. Newman, R. W. Francis, L. H. Gaines and B. M. L. Rao, J. Electrochem. Soc., 127 (1980) 2025.
- 4 P. G. Glugla, in B. B. Owens and N. Margalit (eds.), Proc. Symp. Ambient Temper-

ature Lithium Batteries, Vol. 80-4, The Electrochemical Society, Pennington, NJ, 1981, p. 179.

- 5 G. H. Newman and L. P. Klemann, in H. V. Venkatasetty, (ed.), Proc. Symp. on Lithium Batteries, Vol. 81-4, The Electrochemical Society, Pennington, NJ, 1981, p. 179.
- 6 B. M. L. Rao, D. Eustace and D. Farcasiu, U.S. Pat. 4,284,692 (1981).
- 7 K. M. Abraham, J. Power Sources, 14 (1985) 179.
- 8 L. A. Dominey, J. L. Goldman and V. R. Koch, in A. N. Dey (ed.), Proc. Symp. on Lithium Batteries, Vol. 87-1, The Electrochemical Society, Pennington, 1987, p. 300.
- 9 J. L. Goldman, unpublished results.